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Synthesis of superparamagnetic bare Fe_3O_4 nanostructures and core/shell $(Fe_3O_4/alginate)$ nanocomposites

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ARTICLE INFO

Article history: Received 12 July 2011 Received in revised form 23 February 2012 Accepted 6 April 2012 Available online 13 April 2012

Keywords: Magnetic NPs Composites Biomedical Alginate Core-shell

ABSTRACT

In this article we report about the synthesis of superparamagnetic bare Fe $_3O_4$ nanostructures and core/shell (Fe $_3O_4$ /alginate) nanocomposites by simple low-temperature based method at pH values 5, 9, and 14. The structural morphology and magnetic behavior of Fe $_3O_4$ nanostructures and core/shell (Fe $_3O_4$ /alginate) nanocomposites (Fe $_3O_4$ /alg NCs) have been investigated by X-ray diffractometer (XRD), Fourier transform infrared spectroscopy (FT-IR), Raman spectroscopy (RS), ultraviolet–visible (UV–vis) spectroscopy, transmission electron microscopy (TEM), energy dispersive X-ray spectroscopy (EDX) and vibrating sample magnetometer (VSM). The particle size was calculated by TEM measurements and it turns out to be \sim 10 nm and \sim 14 nm for bare Fe $_3O_4$ nanoparticle and Fe $_3O_4$ /alg NCs with core/shell structure, respectively. The magnetic properties of the synthesized products were found to be function of pH at which the synthesis has been done. The synthesized Fe $_3O_4$ nanoparticle and Fe $_3O_4$ /alg NCs were found to be superparmagnetic in nature at room temperature. We observed that the value of saturation magnetization in case of Fe $_3O_4$ /alg NCs decreases by increasing the pH value.

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1. Introduction

Currently, the investigations on magnetic nanoparticles (MNPs) have attracted considerable attention to the scientist due to its application in various fields such as physics, medicine, biology and materials science. Particularly, superparamagnetic materials play an important role in biomedical applications including magnetic resonance imaging for clinical diagnosis, magnetic drug targeting, hyperthermia anti-cancer strategy (Gupta & Gupta, 2005; Kim et al., 2008; Qu, Liu, Wang, & Hong, 2010) and enzyme immobilization (Singh, Kalita, Singh, & Malhotra, 2011; Singh, Srivastava, Dutta, & Dutta, 2011). Magnetite (Fe₃O₄) and maghemite (γ -Fe₂O₃) are the most commonly used magnetic carriers for a variety of biomedical applications (Liu et al., 2007). Among the various types of magnetic materials, the magnetic–polymer composites represent a class of functional materials where magnetic NPs are embedded in polymer matrixes. Recently, a considerable number of studies

have been done on magnetic-polymer nanocomposites such as; PVP-coated Fe $_3$ O $_4$ (Liu et al., 2007), Fe $_3$ O $_4$ -chitosan and α -Fe $_2$ O $_3$ NPs (Li, Jiang, Huang, Ding, & Chen, 2008; Singh, Kalita, et al., 2011; Singh, Srivastava, et al., 2011), tyrosinase biosensor based on Fe₃O₄ NPs-chitosan nanocomposite (Wang, Bao, Yang, Chen, 2008), barium alginate caged Fe₃O₄/C₁₈ magnetic NPs (Zhang, Shi, Yang, De, & Wang, 2010), polyaniline-coated nano-Fe₃O₄/carbon nanotube composite as the protein digestion (Wang, Tan, Zhao, & Liu, 2008), Fe₃O₄/PPy/P(MAA-co-AAm) trilayered composite (Luo et al., 2010), Fe₃O₄-PVA nanocomposites (Liu, Hu, Liu, Liu, & Chen, 2008), polyaniline/nano-Fe₃O₄/composites (Xiao, Tan, Ji, & Xue, 2007), polymer-Fe₃O₄ nanocomposites (Yang, Brown, Kempel, & Kofinas, 2008), PEI modified Fe₃O₄/Au NPs (Sun, Zhu, Zhang, Zhang, & Wang, 2010), alginic acid-Fe₃O₄/nanocomposite (Unal, Toprak, Durmus, Sözeri, & Baykal, 2010). The biocompatibility of magnetic materials can be tuned by variety of the polymer coating.

Among the various types of polymers, alginate has attracted intense attention as an important class of biomaterial. It exhibits unique properties such as; biocompatibility, inexpensiveness, mild encapsulation process at room temperature and a relatively inert hydrogel environment within the matrix. Moreover, one of the

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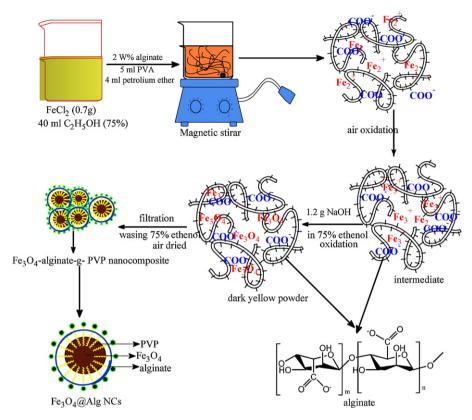


Fig. 1. Schematic diagram show the synthesis method of the Fe₃O₄/alg NCs by in situ co-precipitation method.

unique properties of alginate is its simple gelation with divalent cations like Ca²⁺ which is ionically crosslink carboxylate groups in the G-block of alginate and has been extensively investigated for many biomedical applications such as; cell transplantation matrices, tissue engineering and drug delivery vehicles (Liu et al., 2009). Another imperative approach related to the in situ gel properties of alginate is represented by its possible application in wound dressing because of the presence of calcium ions in the exudates. Alginate can also be mixed with other natural polymers, such as; gelatin and chitosan to produce physical gels. The polymer matrix can be loaded with antibiotics. Alginate can also be chemically modified into low molecular weight oligomers which would be then cross-linked with a biodegradable spacer to form biodegradable hydrogels (Pielesz, 2007).

Alginate is a well-known naturally occurring polysaccharide which is composed of guluronic (G) and mannuronic (M) acid residues (1,4-linked α -L-guluronic acid and β -D-mannuronic acid). This polymer can be regarded as a true block copolymer composed of homopolymeric regions of M (M-block) and G (G-block), interspersed within the regions of alternating structure. It also varies in composition and sequence depending on the source of the alginate. Usually, commercially available alginates are extracted from the cell wall of brown algae. In contrast to algal alginates, bacterial alginates are partially acetylated on the O-2 and O-3 positions of the mannuronate units, which protect them from conversion into guluronate by C-5 epimerase (Schmid, Messmer, Yeo, Zhang, & Zenobi, 2008). This is a possible explanation for their guluronate/mannuronate ratio, which is usually lower than in algal alginates. It has been observed that the physico-chemical properties of alginate are greatly affected by the M/G ratio as well as by the structure of the alternating zones. Alginate-based magnetic composites can be synthesized by incorporating iron oxide particles into calcium alginate hydrogel (Ma, Xiao, Li, Shi, & Zhu, 2008).

Soft magnetic materials with switching properties are prospective for biomedical applications such as; targeted drug delivery and

hyperthermia. However, there are certain limitations because of their toxicity. Moreover, the applications of these nanomaterials are depending on various factors for instance, size of the particles, stability, ambient magnetic properties, and biocompatibility. In fact, the synthesis of magnetic nanoparticles (NPs) with controllable size distribution remains a great challenge for the scientist, particularly, the particle size below 10 nm which show superparamagnetic (switching) properties. Coating of these magnetic nanoparticles with biocompatible polymeric material is the area of intense research. For in vivo usage, these magnetic nanoparticles require a surface modification to ensure their non-toxic and biocompatibility for the body. Considering the hydrophobic and hydrophilic interaction the coating of different types of polymer materials has been reported (Tamhankar, Kulkarni, & Watawe, 2011 and reference therein). An efficient, economic, scalable and non-toxic synthesis of Fe₃O₄ NPs and their composites is highly desired for the potential biomedical applications and fundamental research (Liu et al., 2007).

In this paper, a facile chemical method was developed to prepare aminated-(Fe $_3$ O $_4$ /alginate) core/shell NCs with narrow size distributions using modified co-precipitation method. The characterization of synthesized products was done using XRD, FT-IR, RS, UV-vis, TEM, EDX and VSM techniques. The results show that the synthesized magnetic NPs are superparmagnetic in nature. The method used in the synthesis of magnetic nanoparticle is non-toxic and this approach can be used to synthesize various types of magnetic NPs.

2. Experimental

2.1. Materials

The reagent used for the preparation of Fe₃O₄ NPs and its composites with alginate, were FeCl₂ and FeCl₃·6H₂O, trimethylamine, 75% ethanol, *Laminaria hyperborea* sodium

alginates, provided by Pronova Biopolymers Ltd. (with $M_{\rm W}$ = 60,000–80,000, and M/G = 32/68), petroleum ether, PVP and NaOH of analytical grade, purchased from Merck, New Delhi, and they were used directly without any further purification.

2.2. Synthesis of magnetic Fe₃O₄ NPs

In order to prepare Fe_3O_4 NPs by co-precipitation technique, 0.3 M aqueous solutions of $FeCl_2$ and $FeCl_3 \cdot 6H_2O$ were mixed into the ratio of 1:2. The pH of mixed solution was maintained by mixing the trimethylamine. Two different solutions were prepared for pH values 4 and 10. The mixing of trimethylamine causes the formation of precipitate in both the solutions. The obtained precipitate was filtered and washed several times to remove the impurities of chloride ions. The washed precipitate was drying in vacuum to get the final products.

2.3. Synthesis of Fe₃O₄/alg NCs

Synthesis of Fe $_3$ O $_4$ /alg NCs has been done following the procedure reported in the literature (Ma et al., 2008). Fig. 1 shows the preparation method of Fe $_3$ O $_4$ /alg NCs. In a typical procedure, a mixture of 1 g FeCl $_2$ and 50 mL 75% ethanol is added into the solution of 5 mL 2 wt% alginate–aqueous solutions and subsequently 5 mL petroleum ether was also added under vigorous magnetic stirring. Three samples at different pH (5, 9 and 14) have been synthesized by using an alkaline solution which was prepared by dissolving the 1.2 g of NaOH into 20 mL of 75% ethanol. Subsequently, the mixture is exposed to air at room temperature for 30 min. The resultant material was then filtered and washed for several time with 95% ethanol and further it was dried in air to obtain the final products.

3. Characterizations

3.1. FT-IR

The compositions of the synthesized products were examined by recording the FT-IR spectra in the spectral range of $400-4000\,\mathrm{cm^{-1}}$ using a model of (PerkinElmer, Spectrum BX II) spectrometer.

3.2. XRD

The phase identification and average crystallite size were investigated using powder XRD XPERT-PRO (PW3050/60) equipped with Cu-K α radiation (λ = 1.5406 Å) operated at 30 kV and 30 mA.

3.3. Raman

Raman spectra were recorded in the range of 200–2000 cm⁻¹ at room temperature, using a Renishaw inVia Raman spectrometer equipped with a high-performance CCD detector. The 514 nm line of argon-ion laser was used to illuminate the sample.

3.4. UV-vis

The optical study of the synthesized NPs was done by recording the absorption spectra in the range of 200–800 nm using UV–vis spectrophotometer (Phoenix 2200 DPCV).

3.5. TEM

The particles size of the synthesized product was investigated by HR-TEM (TEM, Tecnai G2 S-Twin) techniques.

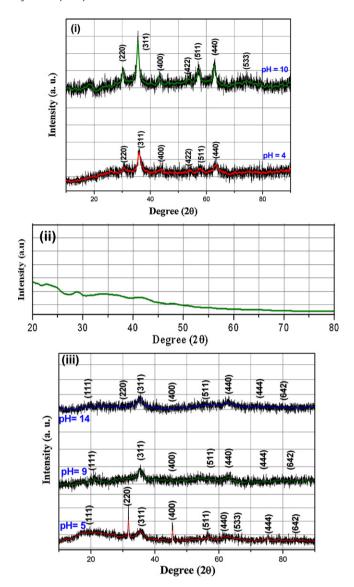


Fig. 2. XRD pattern of (i) Fe_3O_4 NPs (ii) sodium alginate and (iii) Fe_3O_4 /alg NCs synthesized at different pH.

3.6. EDX

Elemental composition was investigated through energy dispersive X-ray spectroscopy (EDX) using (FEI Quanta-200 MK2).

3.7. VSM

The magnetic properties of the synthesized samples were measured at room temperature by the means of VSM (ADE-DMS, model EV-7USA) at the maximum applied field of 17.5 kOe.

4. Results and discussion

4.1. XRD study of the synthesized products

The XRD patterns of the pure Fe_3O_4 NPs synthesized by coprecipitation method are shown in Fig. 2(i). All the diffraction peaks can be indexed to the cubic and inverse type spinel structure of Fe_3O_4 with lattice constant $\sim a = 8.394$ Å, which is in good agreement with the value obtained by JCPDS (card No. 85-1436, a = 8.393 Å). No any impurity phase was detected in the XRD patterns of the synthesized products. The appearance of broad peaks

suggests the nanocrystalline nature of Fe $_3$ O $_4$ particles (Xiao et al., 2007). The average crystallite size has been calculated using Scherrer formula and the values were found to be \sim 10 and \sim 15 nm for the NPs synthesized at pH values 4 and 10, respectively.

The X-ray diffraction pattern of the sodium alginate is shown in Fig. 2(ii). No characteristic diffraction peaks were observed in the spectrum, which implies that a long range disorder along with the amorphous feature exist in sodium alginate polymer (Yu et al., 2006).

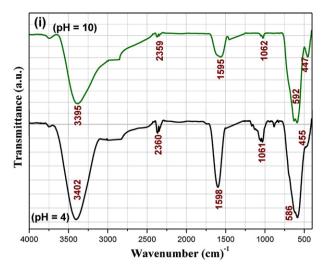
In Fig. 2(iii), the XRD patterns of Fe $_3$ O $_4$ /alg NCs reveal similar characteristic peaks with pure Fe $_3$ O $_4$ nanoparticles, which confirm the formation of Fe $_3$ O $_4$ particles. Comparing to single phase Fe $_3$ O $_4$ NPs, the Fe $_3$ O $_4$ /alg NCs show an amorphous feature by the presence of peak at \sim (2 θ =20°) along with the low intensity of the other peaks. This is attributed to the introduction of amorphous shell of alginate (Fang, Kim, & Choi, 2009). Moreover, it is quite evident from the XRD pattern of the composites synthesized at pH value 5 that it prefers to grow along the direction of (2 2 0) plane. The mean crystallite size of synthesized Fe $_3$ O $_4$ /alg NCs was calculated to be \sim 14 nm.

4.2. FT-IR study of the synthesized products

The FT-IR spectra of Fe_3O_4 NPs and Fe_3O_4 /alg NCs are presented in Fig. 3((i) and (ii)), respectively. Since Fe_3O_4 has inverse type spinel structure and it has been observed that spinel type structure exhibit mainly two absorption bands at $\sim\!600$ and $\sim\!450\,\mathrm{cm}^{-1}$ corresponding to the metal–oxygen band in tetrahedral and octahedral sites. The FT-IR spectra of Fe_3O_4 NPs exhibit two bands around the above said wavenumber positions, which also confirm the formation of spinel type structure. In addition to it, three absorption bands in FT-IR spectra were also observed at $\sim\!3400\,\mathrm{cm}^{-1}$, $\sim\!1600\,\mathrm{cm}^{-1}$ and $\sim\!1060\,\mathrm{cm}^{-1}$ which can be attributed to the stretching, bending and deformation modes, respectively due to the presence of hydroxyl group (Umare et al., 2008).

The molecular structure of alginic acid consists of two carboxylic acid moieties as shown in the synthesis scheme (see Fig. 1). The FT-IR spectra of the alginic acid and PVA functionalized Fe₃O₄/alg NCs synthesized at different pH values are shown in Fig. 3(ii). Addition of Fe₃O₄ in alginate solution shows (spectra at pH 5, 9 and 14) additional peaks in the spectral range 461–595 cm⁻¹. These peaks are attributed to the metal oxide bond (M-O) in alginate matrix. The IR bands of alginate and PVA in the spectral range 3300–3600 cm⁻¹ is due to the stretching modes of OH group. The IR bands of Fe₃O₄/alg NCs become sharper and shifted to the lower wavenumber side compared to pure alginate. This indicates that OH group of alginate and PVA was involved in the assembly of Fe₃O₄ nanoparticles. Thus, M-O-M inorganic network was bonded with alginate macromolecules by hydrogen bonding as well as ionic bonding with Fe₃O₄ nanoparticles for the formation of Fe₃O₄/alg NCs. The band at \sim 1740 cm⁻¹ can be attributed to the free carboxyl group of alginate and the band at \sim 1450 cm $^{-1}$ is assigned to the symmetric COO- stretching vibration. The C=O stretching vibration is missing in the FT-IR spectra of nanocomposites synthesized at pH 9. However, it appears at \sim 1740 cm⁻¹ in the spectrum of pure sodium alginate and nanocomposites synthesized at pH values 5 and 14. The above discussion on the analysis of FT-IR spectra of both, Fe₃O₄ nanoparticles and Fe₃O₄/alg NCs confirms the formation of core/shell Fe₃O₄/alg nanostructures (Sergios et al., 2010).

Two different types of binding possibilities have been suggested for the surface carboxylate bonding. In one possibility, the carboxylate may be connected to the surface through one oxygen atom and both the symmetric C=O and asymmetric stretching were observed. However, in the other possibilities, the carboxylate is bound symmetrically to the surface and only symmetric C=O stretching band



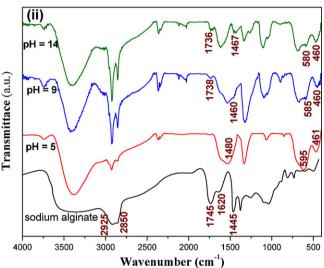


Fig. 3. FT-IR spectra of (i) Fe_3O_4 NPs and (ii) sodium alginate and Fe_3O_4/alg NCs synthesized at different pH.

appears (Unal et al., 2010) in the IR spectra. The difference in the FT-IR spectrum of the nanocomposite synthesized at pH 9 compared to the spectra of nanocomposites synthesized at pH values 5 and 14 may be due the pH dependent structural growth and modifications.

4.3. Raman study of Fe₃O₄ NPs and Fe₃O₄/alg NCs

The Raman spectra of Fe₃O₄ NPs synthesized at two different pH values 4 and 10 have been presented in Fig. 4(i). Raman spectroscopic studies on Fe₃O₄ NPs have been reported by several workers (Degiorgi, Morke, & Wachter, 1987; Gasparov et al., 2000; Gupta, Sood, Metcaff, & Honig, 2002; Shebanova & Lazor, 2003; Soler et al., 2004). In previous studies, there are some discrepancies in the observed number and positions of Raman modes. The Fe₃O₄ has spinel type structure, with 56 atoms in the unit cell and only 14 atoms in the asymmetric unit. As a result, 42 vibrational bands are expected. The analysis of group theory predicts the following vibrational bands (Fateley, Dollish, Devitt, & Bentley, 1997):

$$A_{1g} + E_g + T_{1g} + 3T_{2g} + 2A_{2u} + 2Eu + 5T_{1u} + 2T_{2u}$$

where the T_{1g} , A_{2u} , E_u and T_{2u} vibrational modes are silent in Raman spectrum and thus Fe₃O₄ exhibits only five Raman-active bands with the symmetry combination $(A_{1g} + E_g + 3T_{2g})$ and five

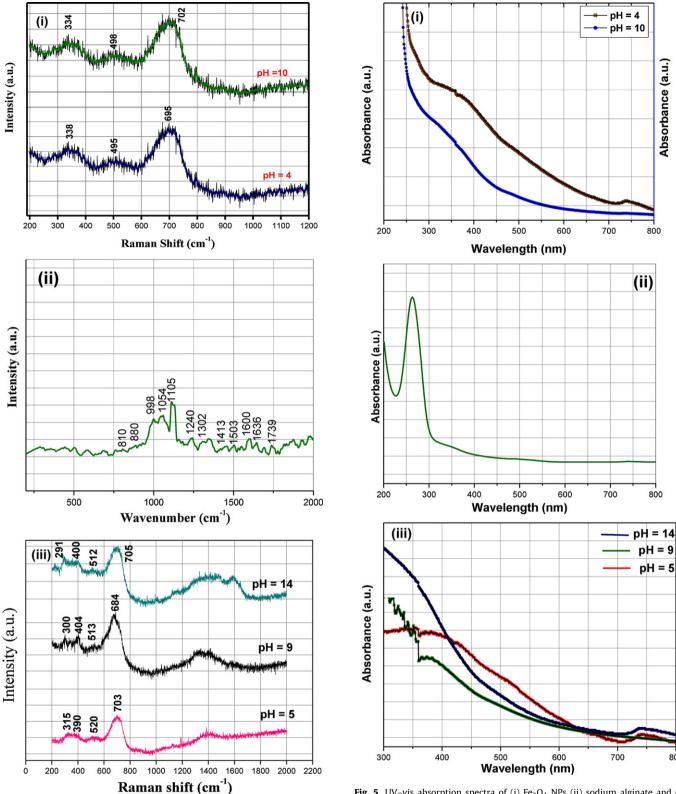


Fig. 4. Raman spectra of (i) $\rm Fe_3O_4$ NPs (ii) sodium alginate and (iii) $\rm Fe_3O_4/alg$ NCs synthesized at different pH.

infrared-active bands $(5T_{1u})$. In the present study, three Raman bands corresponding to the symmetry species E_g (\sim 335), T_{2g} (\sim 495), and A_{1g} (\sim 695) were observed in Raman spectra of the Fe₃O₄ NPs. Fig. 4(ii) represents the Raman spectra of sodium alginate, exhibiting weak Raman vibrational modes. The most

Fig. 5. UV-vis absorption spectra of (i) Fe_3O_4 NPs (ii) sodium alginate and (iii) Fe_3O_4/alg NCs synthesized at different pH.

prominent bands and their assignments according to Raman spectra are shown in Fig. 4(ii). The normal Raman spectra of alginates can be divided into two parts: vibrations of the polymer backbone in the range of <1300 cm $^{-1}$ and stretching vibrations of the carboxylate functional groups with Raman shifts at $\geq \! 1300 \, \text{cm}^{-1}.$ Interactions of alginate with sodium ions lead to band positions

with the most pronounced symmetric and asymmetric COOstretching vibration band at 1413 cm⁻¹ and 1636 cm⁻¹ respectively. The band arises at 1302 cm⁻¹ is due to the stretching vibration of C-O single bond in the G/M unit of the alginate. In the range of glycosidic ring breathing modes at 1105 cm⁻¹ are generally observed, which is attributed to a weakening of C-C and C-O bonds as an indirect consequence of the binding of alginate. The Raman band appears at lower wave number region at 810, 880 and 998 cm⁻¹ corresponding to the skeletal of C-C, C-O stretching, and C-C-H, C-C-O bending modes respectively (Jiron, Leal, Matsuhiro, & Roman, 2009; Schmid et al., 2008; Wagner, Ivleva, Haisch, Niessner, & Horn, 2009). The Raman spectra of the Fe₃O₄/alg NCs are shown in Fig. 4(iii). We observed four Raman bands for Fe₃O₄/alg NCs with small variations in the peak positions compared to the wavenumber position of Raman bands in case of pure Fe₃O₄ nanoparticles. The position and number of observed Raman bands in case of pure Fe₃O₄ NPs and Fe₃O₄/alg NCs were found to be consistent with the earlier study (De Faria, Silva Vena uncio, & de Oliveira, 1997).

4.4. UV–vis absorption study of pure Fe_3O_4 NPs and Fe_3O_4 /alg NCs

The UV–vis absorption spectra of pure Fe_3O_4 NPs, sodium alginate and Fe_3O_4 /alg NCs dispersed in water solution are shown in Fig. 5(i), (ii) and (iii), respectively. The absorption spectra of pure Fe_3O_4 NPs exhibit absorption edge at \sim 470 nm and the optical band gap was calculated to be \sim 2.62 eV. The UV–vis spectrum of sodium alginate is shown in Fig. 5(ii). An intense absorption band at about \sim 265 nm was observed in the spectrum and it can be assigned to double bonds of alginate formed after main chain scission (Yu et al., 2006; Bhattarai & Zhang, 2007). By looking at UV–vis spectra (Fig. 5(i) and (iii)) one can easily notice an obvious difference between the absorbance spectra of pure Fe_3O_4 and Fe_3O_4 /alg NCs. It indicates that Fe_3O_4 nanoparticles interact with the alginate molecules. The observed results were found to be comparable with the earlier study (Zhang, Niu, Cai, & Shi, 2010).

4.5. TEM investigation of Fe₃O₄ NPs and Fe₃O₄/alg NCs

The TEM micrographs of Fe₃O₄ NPs synthesized at pH 10 and 4 are shown in Fig. 6((i), a and b), respectively. The Fe₃O₄ NPs synthesized at pH 10 shows some extent of agglomeration. However, the Fe₃O₄ NPs synthesized at pH 4 shows better dispersion. The average particle size for the particles synthesized at pH 4 was calculated to be ~8.5 nm with help of the distribution curve fitted with Lorentzian function and shown in Fig. 6(c). TEM micrographs of the Fe₃O₄/alg NCs are presented in Fig. 6(ii), (iii) and (iv). TEM images of Fe₃O₄/alg NCs show a better dispersion of the particles in comparison to that of pure Fe₃O₄ NPs and it may be due to the coating of alginate on the surface of the Fe₃O₄ NPs, which essentially reduces the interaction between the magnetic NPs and finally resulting a better dispersion of the nanoparticles. The average size of the nanocomposites synthesized at pH 14 was also calculated by distribution curve and found to be \sim 10.5 nm. The distribution curve is shown in Fig. 6((iv) b). The HR-TEM images of the Fe₃O₄/alg NCs synthesized at different pH exhibit core/shell (Fe₃O₄/alginate) type nanostructures. The SAED patterns of Fe₃O₄/alg NPs exhibit bright rings which suggest the polycrystalline nature of Fe₃O₄/alg NCs (see Fig. 6((ii) a). In order to determine the composition of (Fe₃O₄/alg NCs), synthesized at different pH value EDX, investigations were also performed. Fig. 6(v) shows EDX pattern of the (Fe₃O₄/alg NCs) synthesized at different pH value. The EDX analysis reveal that the wt% of Fe was highest when (Fe₃O₄/alg NCs) was synthesized at pH 5, whereas it was found to be minimum when the synthesis was done at pH 14. The difference in

Table 1 Magnetic properties of the Fe_3O_4 nanoparticles and Fe_3O_4 /alginate nanocomposites.

Sample	Saturation magnetization $M_{\rm s}$ (emu/g)
Fe ₃ O ₄ nanoparticles synthesized at pH 4	31.03
Fe ₃ O ₄ nanoparticles synthesized at pH 10	35.57
$Fe_3O_4/alginate$ nanocomposites synthesized at pH 5	5.40
Fe ₃ O ₄ /alginate nanocomposites synthesized at pH 9	4.69
Fe ₃ O ₄ /alginate nanocomposites synthesized at pH 14	2.35

the composition may due to the response of the chemical reaction at different pH during the synthesis processes, which also influences the degree of polymerization/combination of magnetic nanoparticle.

4.6. Magnetic properties of pure Fe_3O_4 NPs and Fe_3O_4 /alg NCs

The typical magnetization curves of the Fe₃O₄ nanoparticle and Fe₃O₄/alg NCs are shown in Fig. 7(i) and (iii), respectively. The M-H curve was obtained in the presence of magnetic field of 17.5 kOe at the room temperature. In this study, we observed that the Fe₃O₄ nanoparticle and Fe₃O₄/alg NCs exhibit neither coercivity (H_c) nor remanent magnetization (M_r) , which proves that both, Fe₃O₄ nanoparticle and Fe₃O₄/alg NCs are superparamagnetic in nature. Moreover, the M-H curve do not saturate up to the maximum applied field. The values of saturation magnetization (M_s) at the maximum filed are given in Table 1. The reason for superparamagnetic nature of the NPs is essentially due to the very small size. The smaller size (see Fig. 6) may be considered as equivalent to a single magnetic domain where the energy barrier for its spin reversal easily overcomes by thermal vibrations resulting superparamagnetic nature of nano size particles. Further, Morrish and Yu (1955) have reported that Fe₃O₄ NPs having particle size ~50 nm or even less may be considered as a single domain.

The values of M_s of the Fe₃O₄ NPs were found to be 31.03 emu/g and 33.58 emu/g for the Fe₃O₄ nanoparticles synthesized at pH values 4 and 10, respectively. The difference in the values of M_s may be attributed to difference in their size and structural morphology of the Fe₃O₄ NPs. For a comparative study, we also characterized sodium alginate through VSM technique. The M-H loop measured at room (Fig. 7(ii)) temperature exhibit diamagnetic characteristics of the sodium alginate. On the other hand, the values of M_s for Fe₃O₄/alg NCs were found to be decreasing by increasing the pH value (see Table 1). It was reported that the magnetic properties of the composites depend upon the content of Fe₃O₄ NPs (Luo et al., 2010) in the Fe₃O₄/alg matrix. On the basis of above discussion on the pH dependent changes in the $M_{\rm S}$ of the synthesized nanocomposites; we may conclude that by increasing the pH of the solution causes low concentration of Fe₃O₄ NPs in the alginate matrix which essentially reduce the value of M_s of the Fe₃O₄/alg nanocomposites. Thus we found that the change in magnetic properties of the synthesized samples is highly affected due to the change in pH values during the synthesis process, which essentially influence the rate of the chemical reactions causing a significant modification in the structure and composition of the synthesized products consequently the magnetic properties of the products. Decrease in saturation magnetization of the Fe₃O₄/alg nanocomposites is also supported by the EDX results because of decreasing wt% of Fe on

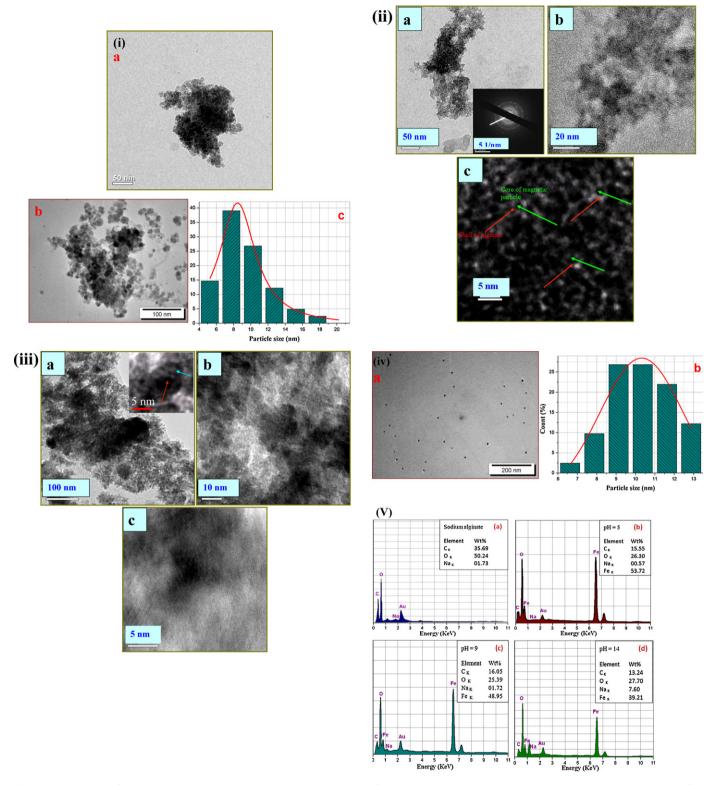
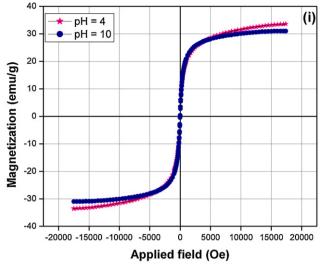
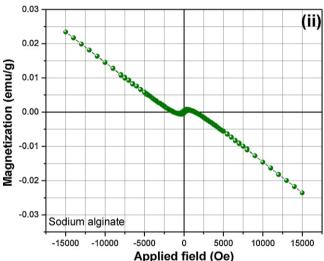


Fig. 6. TEM micrograph of (i) (a) Fe_3O_4 NPs synthesized at pH, 10 (b) TEM micrograph of the nanoparticles synthesized at pH, 4 (c) Particle size distribution curve of the nanoparticles synthesized at pH, 4 (ii) (a-c) Fe_3O_4 and Synthesized at pH, 5 The inset of (ii-a) is the SAED pattern of the nanocomposites (ii-c) exhibit the core of Fe_3O_4 and alginate as the shell (iii) (a-c) is the TEM micrograph of the nanocomposites synthesized at pH, 9. Inset of (iii-a) exhibit the core of Fe_3O_4 (dark image) and alginate as the shell (in light image). Part (iii-c) is the HR-TEM images of the nanocomposites and (iv) (a) TEM micrograph of the Fe_3O_4 alg NCs synthesized at pH, 14 and (v) EDX pattern of sodium alginate (a) and Fe_3O_4 alg NCs synthesized at different pH (b-d).

increasing the pH value. The variations in magnetic properties have been also observed when the products were synthesized at different pH value (Srivastava, Ojha, Chaubey, Sharma, & Pandey, 2010; Wu, Huang, Wang, & Wu, 2005; Zhang, Shi, et al., 2010; Zhang,

Niu, et al., 2010). The synthesized superparamagnetic NPs may be used in various types of biomedical applications because of the superparmagnetic properties (exhibit switching properties), highly desirable for in vivo applications.





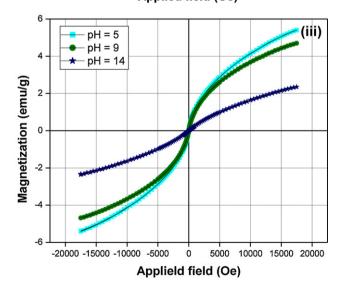


Fig. 7. M–H curve of (i) Fe_3O_4 NPs (ii) sodium alginate and (ii) Fe_3O_4/alg NCs synthesized at different pH.

5. Conclusions

In conclusion, superparamagnetic Fe_3O_4 nanoparticle and Fe_3O_4 /alg NCs were synthesized by co-precipitation and in situ co-precipitation assisted polymerization methods, respectively. TEM micrographs show that Fe_3O_4 NPs were wrapped by a layer of alginate molecules indicating the formation of core/shell (Fe_3O_4 /alginate) type nanostructures. Magnetization measurements show that the synthesized products are superparamagnetic in nature at room temperature. The value of saturation magnetization in the case of Fe_3O_4 /alg NCs decreases by increasing the pH values. The synthesized superparamagnetic NPs and its composite with biocompatible alginate can be used in various types of biomedical applications.

Acknowledgments

MS is thankful to MNNIT, Allahabad for granting the senior research fellowship. JS is thankful to CSIR, India for the award of research associateship. AKO is thankful to the Alexander von Humboldt Stiftung for the award of a research fellowship. The Department of Chemistry of MNNIT, Allahabad is gratefully acknowledged for granting access to the available research facilities. Electron Microscope Facility, Department of Anatomy Institute of Medical Sciences, Banaras Hindu University, Varanasi is highly acknowledged for providing the TEM facility.

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